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Communication

# A 2D covalent organic framework as a high-performance cathode material for lithium-ion batteries

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# ABSTRACT

Organic cathode materials for lithium storage have attracted wide attention owing to their very diverse structures and largely tuned engineered molecular levels. However, it remains a great challenge to design a cathode material with simultaneously combined features of high specific capacity, cycle life and rate performance. Here, based on our proposed strategy, we design and report a BQ1-COF consisting of maximum active groups (C=O and C=N) with minimal inactive groups, which when used as cathode materials for lithium-ion batteries give a reversible capacity of 502.4 mA h g<sup>-1</sup> at 0.05C, so far the highest capacity among polymer-based cathode materials. More importantly, the stable framework structure delivers an excellent capacity retention (81% after 1,000 cycles at 1.54 A g<sup>-1</sup>), and it is noted that the rate performance (170.7 mA h g<sup>-1</sup> even at 7.73 A g<sup>-1</sup>) is far superior to previous related reports. These results indicate that maximizing the loading of redox active groups in a stable network structure is an effective strategy to design organic cathode materials simultaneously with high capacity and outstanding cycle and rate performance for next generation lithium-ion batteries.

High-performance lithium-ion batteries (LIBs) have attracted growing attention for satisfying the demands of communication, portable electronic devices, electric vehicles (EVs) and hybrid EVs (HEVs) [1]. The essential is to have applicable electrode materials with high-performance. In particular, cathode material is probably a bigger bottleneck due to its low theoretical capacity compared to that of anode materials, such as Si-based and Li metals anodes [1,2]. With the advantages of abundant resource, wide structural diversity and being environmentally friendly [3,4], organic/polymeric materials have been explored as electrode materials and in some cases with high capacity [2, 5,6]. However, there have been some intrinsic and challenging issues remained to be solved, and the main one is their dissolution into electrolytes, which leading to a fast capacity fading during cycling. Many strategies have been applied to address such an issue, including polymerizing the redox-active compound [7–9], elongating the molecular conjugation length [10,11], forming a salt with organic carbonyl

compounds and so on [12–16]. Nevertheless, while these strategies succeed in enhancing the cyclability, they come with the cost of reduced specific capacity due to a large amount of existence of inactive units. The inherent low conductivity and redox voltage for organic materials also limit the electrochemical performance of their LIBs [17]. Thus, it is still a great challenge to design appropriate cathode materials with high capacity, outstanding cyclability and rate performance all together.

For the capacity issue, on the basis of the relationship between theoretical capacity and chemical structure [18,19], the most intrinsic approach to achieve the highest theoretical capacity is to increase redox group loading and reduce inactive group loading as much as possible, thus with the maximum electron transfer per repeating unit. In theory, the compounds that consist of only redox groups without any inactive groups could deliver the highest capacity. This comes to the case such as (CO)<sub>6</sub>, with only redox active C=O group [19], exhibiting the highest theoretical capacity (957 mA h  $g^{-1}$ ) among conjugated carbonyl

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compounds. Nevertheless, its cycle and rate performance need to be further improved for practical application. Clearly, the most effective way to obtain high capacity and long cycle life simultaneously is to maximize the loading of redox active groups in a stable polymeric network structure to have monolithic materials.

With this, we propose a series of possible materials that meet the above requirements with the most widely used redox groups (C=N, C=O), as shown in Scheme 1 and Table S1, including linear polymers and covalent organic frameworks (COFs). In particular, COFs exhibit more distinct advantages as electrodes materials for LIBs due to its rather stable framework structure and regular porous channel, which should facilitate both ion and electron transport for both better cycling stability and rate performance [20–26]. Therefore, the designed COFs (Scheme 1) with the maximum electron transfer per repeating unit are expected to not only deliver a high capacity but also afford outstanding cycle and rate performance for LIBs.

Among above proposed structures, considering both the structural stability for cycling issue and theoretical capacity together with easy access, BO1-COF was chosen and synthesized directly by a simple polycondensation reaction of hexaketocyclohexane (HKH) and tetramino-benzoquinone (TABQ) [25] as shown in Fig. 1a and more details are given in the Methods. The abundant C=O and C=N groups and minimal inactive groups in this structure render it a theoretical capacity of 773 mA h  $g^{-1}$  corresponding to 18 electrons redox per repeating molecular unit. When applied as the cathode material for LIBs, it delivers an ultrahigh experimental capacity of 502.4 mA h  $g^{-1}$  corresponding to a maximum number of 12 electrons transfer per repeating molecular unit, so far the highest capacity among polymer-based cathode materials. The stable framework structure affords an excellent capacity retention of 81% after 1,000 cycles at 1.54 A  $g^{-1}$ . The well-defined channel and extended conjugated system promote a rapid Li<sup>+</sup> diffusion and electron transport, delivering an outstanding rate performance (170.7 mA h  $g^{-1}$  at 10C), which is far superior to previous related reports.

FT-IR and solid-state <sup>13</sup>C NMR were carried out to confirm the chemical composition and structure of BQ1-COF (Figs. S1 and S2). The emerging C=N group (1550 cm<sup>-1</sup>) in the FTIR spectrum of BQ1-COF and the peak with a chemical shift of around 146 ppm in the solid-state <sup>13</sup>C NMR spectrum, indicating the formation of the expected  $\pi$ -conjugated C=N linkages, consistent with the structures proposed in Fig. 1a. The chemical composition and structure of BQ1-COF were also characterized and supported by X-ray photoelectron spectroscopy (XPS) (Fig. S3) and Elemental analysis (Table S2).

As shown in Fig. 1c, the Brunauer-Emmett-Teller (BET) surface area was measured by N<sub>2</sub> adsorption-desorption measurement and is calculated to be 94.73 m<sup>2</sup> g<sup>-1</sup> in the range of P/P<sub>0</sub> = 0.05–0.2. Its isotherm

## **Our strategy**

plots exhibit a classic type I isotherm, revealing the microporous character of BQ1-COF [20]. The pore size distribution demonstrates that BQ1-COF forms a well-defined channel structure, and the dominant pore size observed at 1.35 nm is close to the calculated pore size (1.39 nm) as shown in Fig. 1a. And Energy-dispersive X-ray spectroscopy (EDS) elemental mapping (Fig. S5) from the SEM images show the homogeneous spatial distributions of C, N, O of BQ1-COF, and the atomic ratio of C to N is about 2.33, which agrees with the theoretical calculated value of 2.5.

Its XRD pattern (Fig. S6) exhibits clear peaks at 11.88°, 15.56°,  $18.06^{\circ}$  and  $19.92^{\circ}$ , which are assigned to the (110), (101), (210) and (201) planes, respectively, indicating a good crystallinity. The PXRD diffraction peaks for BQ1-COF is consistent with the result of simulated AB stacking structure (Fig. S7). It is worth noting that BQ1-COF does not show a diffraction peak of (100) instead of exhibiting a strong diffraction peak at 28.80°, which is assigned to the (002) plane representing a  $\pi$ -stacking distance of 0.315 nm. The result is probably induced by the strong  $\pi$ - $\pi$  interaction between neighbor layers of BQ1-COF with extended  $\pi$ -conjugated structure [27,28]. In addition, in high-resolution transmission electron microscopy (HRTEM) images of BO1-COF (Fig. 1d, e and S8), the clearly observed lattice fringes with an estimated d spacing of 0.23 nm is close to the inter-spacing of (420) plane (0.24 nm). The observed hexagonal electron diffraction (SAED) pattern (inset in Fig. 1d) is also in consistent with the observed (420) plane in HRTEM. All these indicate a good crystallinity of BQ1-COF.

Furthermore, its Raman spectrum (Fig. S9) exhibits a band at ~1520 cm<sup>-1</sup>, reflecting the in-plane bond-stretching vibrations of sp<sup>2</sup> carbon atoms in the 2D honeycomb structure [29], which further confirms the graphene-like 2D structure of BQ1-COF. Its high thermal stability was demonstrated in the TGA analysis (Fig. S10), which indicates the material is stable up to ~360 °C without clear degrading/decomposition. The lowest unoccupied molecular orbital (LUMO) energy level of BQ1-COF (Fig. S11) (-4.20 eV) is lower than most organic materials, which implies larger electron affinity and higher reduction potential and thus a high working voltage [30–32]. And compared to other reported COFs [33,34], the LUMO-HOMO gap (3.41 eV) is much smaller, indicating possibly a high electrical conductivity [30]. This was consistent with the observed conductivity of  $3.1 \times 10^{-6}$  S m<sup>-1</sup> (Fig. S12), several orders higher than most organic materials [31].

The electrochemical performance of BQ1-COF was evaluated as the cathode material for LIBs with metal lithium as the counter/reference electrode. Fig. 2a shows that the typical cyclic voltammograms (CV) of BQ1-COF electrodes in the initial three-cycle, displaying three clear reduction peaks at 2.64, 2.20 and 1.2 V vs. Li<sup>+</sup>/Li, which clearly point to the multiple-electron redox reactions during the charging/discharging processes. Correspondingly, three clear discharge plateaus at about



Scheme 1. Some proposed organic cathode materials with ultrahigh theoretical capacity with typical redox active groups (C=N, C=O). The structural repeating units are marked by red. The chemical formulae of repeating units and calculated theoretical capacity of each structure are shown below the corresponding structures.



Fig. 1. (a) Scheme for the synthesis of BQ1-COF. (b) Graphic representation of one-layer-conjugated skeleton of BQ1-COF. (c) Pore-size distribution and N<sub>2</sub> adsorption-desorption isotherms of BQ1-COF. (d, e) Transmission electron microscopy images of BQ1-COF. Inset: SAED pattern of BQ1-COF.



**Fig. 2.** (a) The initial three-cycle CV curves of BQ1-COF at a scan rate of  $0.5 \text{ mV s}^{-1}$ . (b) The initial three-cycle capacity-voltage profile of BQ1-COF at a current of 0.05C. (c) Rate capabilities of BQ1-COF at various current rates. (d) Nyquist plots of BQ1-COF before and after 300 cycles at a current of 2C. (e) Long-term cycling performance of BQ1-COF at a current of 2C for 1,000 cycles.

3.20–2.47 V, 2.47–2.11 V and 1.4–1.2V can be distinguished from the fully reversible galvanostatic charge/discharge curves in Fig. 2b. The first cycle CV curve shows an extra anodic peak at 3.36 V compared to the next cycles, which is due to the formation of the SEI film [10,35]. The second and third CV curves display a good reversibility of the redox process. In Fig. 2b, the first galvanostatic charge/discharge curve shows a high initial capacity of 494 mA h g<sup>-1</sup> with a high coulombic efficiency of 98.4%, and the charge/discharge curves of the second and third cycles are also consistent each other. These results indicate the highly

reversible nature of BQ1-COF as cathode material for LIBs. It's worth of noting that the cathode exhibits an ultrahigh capacity of 502.4 mA h g<sup>-1</sup> in second cycle at 0.05C. To our knowledge, the reversible capacity of BQ1-COF is the highest among all reported organic COF or polymer-based cathode materials (Fig. 4c, Table S3) [7–9,15,20–23,25, 34,36–38]. Furthermore, BQ1-COF exhibits an average discharge voltage of 2.06 V, and the energy density, calculated based on the reversible capacity and average discharge voltage above, can reach up to 1033 W h kg<sup>-1</sup>, much higher than the capacity (100–140 W h kg<sup>-1</sup>) of

traditional commercial cathode materials such as  $LiCoO_2$  and  $LiMn_2O_4$ [31]. Even at a higher current density of 10C (7.73 A g<sup>-1</sup>), the cathode material can still afford a high energy density (~350 W h kg<sup>-1</sup>) with also a high power density (12.6 kW kg<sup>-1</sup>) (Figs. S13 and S14).

The rate capabilities of BQ1-COF (Fig. 2c) were investigated at various current density. Notably, BQ1-COF can still show a reversible specific capacity of 170.7 mA h g<sup>-1</sup> even at an ultrahigh current density of 10C (7.73 A g<sup>-1</sup>). The rate capabilities are far superior to previous related reports (Fig. 4d, Table S4) [7–9,20–23,25,30,34,36–38], which is contributed to its well-defined channel structure and extended  $\pi$ -conjugated systems. To investigate the kinetics of BQ1-COF, electrochemical impedance spectroscopy (EIS) of BQ1-COF was measured, before and after 300 cycles at 2C (Fig. 2d). The charge-transfer resistance (R<sub>ct</sub>) of the cathode material (23.3  $\Omega$  before cycling) decreases to be 20  $\Omega$  after 300 cycles, suggesting a fast and stable reaction kinetic of the cathode [20,39].

In addition, BQ1-COF exhibited outstanding long-term cycling stability (Fig. 2e), while maintaining 198.4 mA h g<sup>-1</sup> with 81% capacity retention after 1,000 cycles at a current density of 2C, and the Coulombic efficiency remains approximately 100% during all the cycling. Furthermore, even at higher current density of 5C (3.87 A g<sup>-1</sup>), BQ1-COF can still deliver capacity of 158 mA h g<sup>-1</sup> after 1,500 cycles, (Fig. S15), revealing an excellent electrochemical stability of BQ1-COF.

To explore the lithium storage mechanism of BQ1-COF, DFT and ex situ XPS were carried out. For the extended conjugated system constructed by C=O groups and neighboring C=N groups, N and O atoms work as the collaborative centers during the lithium intercalation, which can form a stable five-membered chelate rings (C–N–Li–O–C) [40–42]. In addition, electroactive pyridine N and pyrazine N atoms in a  $\pi$ -conjugated C=N system can strongly coordinate with lithium ion, affording high-capacity lithium storage with multiple electron redox reaction [31, 41]. In the initial stage, there are three possible ways of lithiation, the two possible simultaneous coordination of N and O to Li (N–Li–O) and the similar chelating effect of N–Li–N. The DFT calculations for three situations are carried out and the optimization results (Fig. S16) indicated that lithium ions are expected to firstly chelate with oxygen atom and adjacent interior nitrogen atoms to form the configuration of five-membered ring (C-N-Li-O-C). Considering the structural symmetry, three lithium ions could be chelated at the same time in each stage. In the second stage, three possible configurations can be obtained as shown in Fig. S16, together with their calculated total energy, which suggest that lithium ions prefer to coordinate with the nitrogen and oxygen of the same side at this stage. For the next two stages, there are two possibilities for COF-9Li (Fig. S16) and only configuration for COF-12Li (Fig. S17), where six lithium ions might correlate with the COF via chelating with the other three carbonyl groups and six neighboring conjugated C=N groups as observed before [40]. At the final two stages of the lithiation processes, the lithium storage should be mainly contributed to the remaining six electroactive C=N groups of the COF-12Li. As indicated by Fig. S17, COF-15Li should be formed by accepting three lithium ions that bond with three pyridine nitrogen atoms. Afterwards, with the insertion of another three lithium ions, the final lithiated structure of COF-18Li can be achieved. Therefore, one repeating unit of BQ1-COF that contains 6C=O groups and 12C=N groups can reversibly accept 18 lithium ions in theory, making full advantage of redox groups (C=O, C=N) and offering an ultrahigh theoretical capacity of 773 mA h  $g^{-1}$ .

Nevertheless, it is also noted that the binding energies ( $\Delta E$ ) elevate significantly after inserting 12 lithium ions (Table S5), so the following lithiation will be much more difficult. This can be explained by the steric hindrance effect and decreased stability of the produced molecule after 18 Li<sup>+</sup> insertion as shown in Figure S17 [40]. Therefore, the stepwise actual lithium-storage process of BQ1-COF can be divided into four stages (Fig. 3), and a maximum of 12 lithium ions can be stored experimentally in a COF repeating unit in the operating voltage window of 3.5–1.2V. The total energy of BQ1-COF and lithiated COF-nLi (n = 3, 6, 9, 12) at each stage are marked by orange in Fig. S18, the trend of energy change is well consistent with the discharge curve.

Ex situ XPS scans were performed to identify the structural evolution of BQ1-COF during its discharging/charging process. As shown in



Fig. 3. Scheme for the structure revolution during the 12 lithium ions insertion and extraction process with a COF repeating unit.



Fig. 4. (a) Ex-situ XPS local scan spectra of O 1s and (b) N 1s regions at different states during lithiation and delithiation process, respectively. (c) Theoretical and measured capacities and (d) rate performance comparison of BQ1-COF with some previously reported COF and related polymer-based cathode materials, respectively.

Fig. 4a, the peak for O 1s spectrum shifted toward lower binding energy during the lithiation process and then recovered to the initial state during the following delithiation process. This indicates that the C=O groups of the cathode participate in the reversible redox reaction with lithium ions [41-43]. As shown in Fig. 4b, for the as-prepared cathode, the N1s spectrum could be divided into two peaks located at 399.17 and 400.5 eV, which are assigned to the conjugated  $(sp^2)$  pyrazine and non-conjugated (sp<sup>3</sup>) amino-groups, respectively. After lithiation, the intensity of C=N bond decreased, while an additional peak evolved at a lower binding energy of 398.0 eV, corresponding to the formation of a new bond (N-Li) between N and Li atoms [41,43], which implies that C=N groups are also redox-active centers similar to C=O groups. For the delithiation cathodes, the restoration of C=N peak was observed, while the N-Li peak was disappeared, suggesting the reversible Li<sup>+</sup> insertion C=N bonds during the redox process. These ex situ XPS results confirm that there happens reversible change involving both C=O and C=N groups in the molecular structure of the cathode during the redox reaction.

In summary, a 2D BQ1-COF with multiple redox sites was designed and prepared as the cathode material for LIBs and exhibited an excellent overall battery performance. The observed ultrahigh reversible capacity is contributed to the abundant active sites associated with not only C=O but also  $\pi$ -conjugated C=N groups, delivering the highest capacity and ultrahigh energy density among polymer-based cathode materials. Moreover, the stable 2D conjugated network of BQ1-COF avoids the common dissolution of small molecular-based organic cathode materials and endows this cathode material with excellent cycling life. In addition, the well-defined channel and extended conjugated framework can promote the rapid Li<sup>+</sup> diffusion and electron transport, leading to a prominent rate performance. Our design strategy is expected to offer one way towards the development of high performance organic cathode materials for next generation LIBs with high specific capacity and outstanding cycling stability as well as rate performance all together.

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#### Appendix A. Supplementary data

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